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Composite Material

The invention relates to a composite material of a substrate with, applied to at least one side, a titanium oxide layer with a chemical, physical, mechanical, catalytic and/or optical function. The invention further relates to a process for the production and uses of the composite material.

The broad term substrate comprises firstly easily flammable and/or heat-sensitive materials of polymers, polymer-like or natural materials, but also materials of metal, glass, ceramic and combinations thereof (composites), for which a coating process at low temperatures is preferred for process technical reasons. The substrates are coated with ceramic titanium oxide layers which fulfil one or more protective effects and for example thus increase safety in everyday dealings with highly flammable and/or easily contaminated materials.

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The burning behaviour of materials (in particular textiles, films and plastic containers) and the assessment of their fire risk is of great importance as they are always present in the human environment. Textiles are used for example in a multiplicity of applications mainly as clothing, domestic textiles and technical textiles. The combustion process is initiated by way of heating, decomposition and combustion of the flammable material. Depending on its composition, under the effect of heat the material will first melt, flow or remain unchanged, and on further supply of energy will finally decompose and hence develop heat. After ignition of the flammable material the flames are propagated by way of its decomposed surface, where the speed of flame propagation is accompanied by heat emission from the material. As well as combustibility therefore flame propagation and the degree of heat emission are parameters which determine fire.

The required flame protection can be achieved traditionally in various ways. Firstly intrinsically flame-protected polymers can be used such as polyvinyl chloride (PVC) or fluoropolymers. Combustible polymers such as polyethylene (PE), polypropylene (PP) or polyamide (PA) can be given various flame-inhibiting additives (e.g. aluminium hydroxide, magnesium hydroxide, organic bromine

compounds). Usually however a high proportion of these additives in the polymer matrix is required to achieve adequate flame protection. This is expressed in a high density, loss of flexibility and low mechanical properties of the material.

Titanium dioxide (TiO₂) has known properties as a photo-semiconductor, a high refractive index, a high transparency in the visible and near-infrared wavelength range, a high dielectric constant, gives very good wear protection, is chemically inert and finally has excellent thermal properties. TiO₂ crystallises in three modifications: tetragonal rutile, anatase and orthorhombic brookite. Special experimental conditions are required to be able to produce brookite. Rutile is interesting for optical applications while the photo-catalytic properties of anatase are more pronounced due to the optical band gap at 3.2 eV.

There are numerous sub-oxides of titanium (TiO_x) with an oxygen content of $0.7 \le x < 2$. TiO_x with an oxygen content of 0.7 < x < 1.5 at room temperature has an electrical resistance of around 400 $\mu\Omega$ cm, at a higher oxygen content this increases rapidly, and TiO₂ is an insulator.

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It is known that with TiO₂ layers, the crystallinity and its modifications depend on the production method, process parameters and coating configuration. Usually crystalline TiO₂ layers are produced with sol-gel processes, spray pyrolysis, painting, electron beam vaporisation on metal-organic chemical vapour deposition (MOCVD) above 300°C. TiO₂ layers which are produced with reactive vaporisation or plasma- activated chemical vapour deposition methods (PACVD) below 300°C are usually amorphous and less dense. If these amorphous layers are tempered between 300° and 500°C, the anatase structure of TiO₂ is dominant; on heat treatment above 600°C, the TiO₂ modification rutile forms.

Secondly, amorphous or crystalline TiO_x or TiO₂ layers can be generated below 300°C with methods which are characterised by a higher particle energy - e.g. reactive or non-reactive magnetron sputtering (cathode sputtering), non-filtered or filtered spark discharge, ion-beam-assisted deposition (IAD) and pulsed laser

deposition. With RF sputtering, depending on the choice of coating parameters, TiO₂ can be deposited amorphous or crystalline on an unheated material.

In SURFACE AND COATINGS TECHNOLOGY 102 (1998), 67 - 72, thin titanium dioxide layers are described which are deposited by RF sputtering in an argonoxygen atmosphere. The microstructures of the TiO₂ deposition vary within a broad range from compact to porous and columnar. The O/Ti ratio rises as the pressure increases when the other reaction parameters remain unchanged. The publication deals primarily with scientific experiments.

The invention is based on the object of creating a composite material and a process for its production with a functional titanium oxide layer of the type cited initially which brings improved, in particular synergetic functionalities for a wide range of substrates. An interaction of oxygen and other reactive gases with the substrate should be prevented and said substrate isolated thermally.

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With relation to the composite material, the object according to the invention is achieved in that on the substrate is deposited a titanium oxide layer of a base layer of TiO_x with an oxygen content of $0.7 \le x < 2$ or $TiO_x(OH)_y$ with an oxygen content of $0.5 \le x < 2$ and a hydroxide content of $0 \le y < 0.5$ and on this base layer is applied a top layer of amorphous and/or crystalline TiO_2 . Special and refined embodiments of the composite material are the subject of dependent claims.

The substrate with a base layer and a top layer, where applicable also with further layers, is referred to here and in general for the sake of simplicity as a composite material. Furthermore to avoid repetition, the term TiO_x also always includes the variants TiO_x(OH)_y. The terms TiO_x, TiO_x(OH)_y and TiO₂ comprise pure titanium oxide layers but also titanium oxide layers with other metal oxides where the base layer as a whole contains less than 50 w.%, the top layer is as a whole less than 7 w.% of other metal oxides listed in detail below.

The titanium oxide layer according to the invention is a multi-functional layer which protects a substrate e.g. from combustion, contamination, degradation

(migration of additives, photo-oxidization). This allows any material to be given flame protection, a hygienic protection (self-cleaning, germicidal effect), anti-static protection and/or an anti-fogging effect. Such a composite material is suitable for example for use in the medical sector, for household accessories, domestic articles, textiles, carpets, cables and photovoltaics, and in cleaning plants for water, watery solutions and air.

Suitable materials to be protected are in particular highly flammable and/or heatsensitive materials such as polymers, low melting metals, composites and natural substances in the form of rigid to flexible films, fabrics, membranes, fibres, tubes, plates, containers and powders.

The titanium oxide layer preferably has a total layer thickness of 3 to 1000 nm, where the top layer comprises at least around 10% of the total layer thickness. The top layer comprises titanium dioxide, TiO₂, in practice however the transition is flowing and a value of TiO_{1.99} for example can be allocated to the top layer. Furthermore in practice ultra-thin layers of just 3 nm occur rather rarely, suitably the entire layer thickness is in the range of 10 to 300 nm, in particular 20 to 150 nm, where 10 to 50% of the entire layer thickness consists of the top layer.

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On use of substrates of plastic and natural substances (in particular wool and cotton), a titanium dioxide layer can be problematical, it can also as a catalyst triggering a decomposition of the substrate surface. With plastics and natural substances it may be suitable, before application of the base layer of TiO_x , to apply a protective layer of at least one metal oxide of the group which preferably comprises MgO, ZnO_x , In_2O_x , In_2O_x , Sb_2O_x , Al_2O_x and SiO_x , and/or a polar adhesion layer as an adhesion-promotion layer. The choice of optimum metal oxide or optimum mixture of metal oxides can easily be determined by the specialist by experiment. For a base layer of TiO_x with an oxygen content x < 1.9 and/or a significant hydroxide content of 0.2 < y < 0.7, there is usually no danger for the substrate.

In a further variant of the titanium oxide layer, between the base layer and the top

layer can be deposited an electrically conductive intermediate layer which preferably comprises TiO_x with an oxygen content of $0.5 \le x < 1.5$. The electrical conductivity diminishes above an oxygen content of $x \ge 1.5$. The layer can no longer be regarded as electrically conductive, a top layer of TiO_2 with an oxygen content of x = 2 is an insulator. Clearly, an electrically conductive intermediate layer can be deposited in particular when the oxygen content of the base layer lies above x = 1.5 and if an anti-static effect is to be achieved.

As will be explained in more detail later, at least the top nine atomic layers of the top layer mainly comprise the crystalline TiO_x modification anatase, which corresponds to a layer thickness of around 3 nm.

When the multi-functional titanium oxide layer is used as a flame protection layer of a plastic substrate, sub-micron filler particles of a metal oxide can be added, for example TiO_x and/or Sb_2O_3 , or a metal hydroxide which dehydrates under heat, for example $AI(OH)_3$ and/or $Sb(OH)_3$. In this case the TiO_x base layer suitably has an oxygen content of $1.5 \le x \le 1.9$.

In relation to the process for deposition on a substrate of a titanium oxide layer with a chemical, physical, mechanical, catalytic and/or optical function, the object according to the invention is achieved in a first variant in that first is deposited a reactive base layer of TiO_x with an oxygen content of $0.7 \le x < 2$, then by increasing the oxygen content, process pressure, power and/or substrate temperature a top layer of an amorphous or crystalline TiO_2 is deposited.

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In the second variant the object is achieved in relation to the process for depositing on a substrate a titanium oxide layer with a chemical, physical, mechanical, catalytic and/or optical function in that first reactively or non-reactively a base layer is deposited of TiO_x with an oxygen content of $0.7 \le x < 2$ and then electrochemically, thermally and/or with a plasma process the surface is post-oxidized until the base layer is restructured at least partly into a top layer of amorphous or crystalline TiO_2 .

After both processes a top layer of TiO₂ is produced. The process parameters are set so that the top layer usually constitutes at least 10% of the total layer thickness. For extremely thin layers according to the second variant the entire base layer can be restructured into a TiO₂ layer, but this is not usually the case.

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The application takes place with the methods which are known in themselves and already mentioned above, for process technical reasons coating processes at low temperatures are preferred. Any intermediate layer between the base and cover layer and a protective layer between the substrate and the base layer are also deposited using one of the said methods.

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Preferably, in particular with a plastic substrate or non-polar material, the base layer or protective layer is applied after plasma activation of the substrate surface. This increases the adhesion of the layer to be deposited. Pretreatment can also take place by means of an ultra-thin polar plasma layer of a few nanometres thickness. This polar plasma layer firstly increases the adhesion of the base layer and secondly prevents degradation of the substrate. For the generation of a polar layer with long-term stability, reference is made to WO 99/39842, according to which for polar coating a water-free process gas is used which contains at least one also substituted hydrocarbon compound with up to 8 C/atoms and an inorganic gas.

The ceramic coating can take place directly after the surface treatment of the substrate or later.

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In a refinement of the process a base layer of TiO_x mixed with at least one metal oxide can be deposited. Suitable metal oxides are for example MgO, ZnO, ZrO₂, In₂O₃, Sb₂O₃, Al₂O₃ and/or SiO₂, where the proportion of TiO_x after mixing remains above 50 w.%. Furthermore, the top layer of TiO₂ can also be doped with Fe₂O₃, WO₃, MnO₂, NiO, BaO and/or CaO, where the proportion of TiO₂ after doping remains above 93 w.%. If metal oxides of both groups are added to the base layer of TiO_x, the total proportion of all metal oxides must remain below 50 w.%, the proportion of the added doping metal oxides of the second group must

remain below 7 w.%.

The deposition according to the invention of a base layer of TiO_x (0.7 $\le x < 2$) and a top layer of TiO_2 brings numerous advantages listed inexhaustively below :

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 The process can be performed at a substrate temperature of ≤ 200°C which is important in particular for polymer substrates. Also a low temperature process may be indicated for metals, ceramics and composites and combinations thereof.

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- Deposition of an electrically conductive TiO_x layer onto electrically nonconductive substrates reduces the electrostatic charge and thus supports the hygiene protection synergetically.
- The coating of an organic chemical substrate (polymer, natural substance) with a base layer of TiO_x (x < 1.9) and/or TiO_x(OH)_y with a significant hydroxide proportion is usually non-problematical (no degradation).
- A thin coating has the advantage that the mechanical and processing properties of the substrate are retained. This is particularly important for the processing of fibres and films which must subsequently withstand further treatment processes.
 - With a plasma-activated process, prespecified layer properties such as porosity, crystallinity, density, electrical conductivity, refractive index and polarity can be produced in a targeted manner. In particular, the combination of dense with porous nano-structured multilayers which can have different electrical conductivity and refractive indices, can achieve a synergetic functionality of the titanium oxide layer. For example, the topography of the substrate can be changed or supplemented with suitable layer topography such that this synergetically reinforces the cleaning and hygiene functions.
 - The synergetic multi-functionality of the titanium oxide layers can be adapted

to the application concerned. The plasma-activated low temperature process preferred for production of the layer systems e.g. by magnetron sputtering, spark discharge and plasma MOCVD, are particularly suitable for varying the stoichiometry and layer structure by means of simple process management, and for stabilising the modification anatase by means of the doping of a titanium oxide layer with at least one metal oxide, for example Fe₂O₃, which is easy to perform process technologically. Low temperature processes are therefore also interesting for materials which are not sensitive to heat, such as glass.

Thanks to a titanium oxide layer according to the invention with a base layer of TiO_x and a top layer of TiO_2 which has a thickness of > 3 nm, in particular > 10 nm, hygiene protection, biocompatibility, anti-fogging effect and hence active flame protection can be achieved on practically all substrates. Thanks to the underlying TiO_x base layer biocompatibility, degradation protection of the substrate, passive flame protection, anti-static effect, migration and diffusion barrier protection are also guaranteed.

A photocatalytically active hygiene protection layer of TiO₂ has the ability, in damp atmospheres and under daylight or UV radiation, to decompose various organic compounds on the surface (compounds containing carbon and/or nitrogen, such as oil, bacteria). Thanks to the reduction in contact angle between water and the TiO₂ surface of the top layer, the result is also an anti-fogging effect and a supported removal of dust particles. This hygiene protection, also known as a self-cleaning effect, synergetically reinforces the passive flame protection of a flammable substrate. In this case there is active and reactive flame protection.

In passive flame protection the direct contact of the atmosphere with highly flammable substrate is reduced by the coating according to the invention with a thermally stable titanium oxide which on fire forms a crust of heated titanium oxide. The speed of flame propagation is reduced by the crust and the development of gases escaping from the substrate is reduced (diffusion barrier), which finally can lead to extinction of the flame by passive protection.

The invention is explained in more detail with reference to embodiment examples shown in the drawing which also the subject of dependent claims. The mostly partial cross sections depict diagrammatically:

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- Fig. 1 a film-like composite material with a titanium oxide layer deposited on one side,
- Fig. 2 a variant according to fig. 1 with a two part titanium oxide layer,
- a variant of fig. 2 with a titanium oxide layer deposited on both sides, - Fig. 3
- Fig. 4 10 a fibre with a three part titanium oxide layer, and
 - Fig. 5 a variant according to fig. 2 with an additional protective layer.

Fig. 1 shows a composite material 10 with a substrate 1 and applied on one side a titanium oxide layer 2 without further specification. Figure 1 corresponds to the 15 usual prior art, a titanium layer 2 is applied to a substrate 1 where it fulfils a protective or other function. Fig. 1 however shows also a special case of the present invention. A thin TiO_x layer has been applied which is post-oxidized into TiO₂. Because of the extremely thin layer, the TiO_x layer has been oxidized over the entire thickness into TiO₂. The substrate 1 which is shown merely partially can e.g. be a film, fabric, membrane, plate, fibre, tube, cable or container part and comprise a conventional material.

In Fig. 2 the titanium oxide layer 2 is divided into a base layer 3 of TiO_x where the oxygen content is $0.7 \le x < 0.2$, and a top layer 4 of TiO₂. Distributed finely dispersed in the substrate 1 are submicron particles 6 of a metal oxide/metal hydroxide. The top layer 4 of TiO₂ is mainly present in the tetragonal crystal structure anatase.

The transition from the base layer 3 to the top layer 4 is shown sharply. If the base layer 3 is partly converted into a top layer 4 by means of post-oxidization, the 30 transition is flowing.

Fig. 3 shows a composite material 10 with titanium oxide layer 2 applied to both

sides. The structure of this titanium oxide layer 2 corresponds to that in fig. 2.

In fig. 4 the substrate 1 is a textile fibre and deposited directly on the base layer 3 is an electrically conductive intermediate layer 5 which surrounds as a cylinder casing the base layer 3 which is deposited directly on the fibre. This electrically conductive intermediate layer 5 comprises TiO_x and has an oxygen content of 0.7 < x < 1.5. Above the intermediate layer lies the top layer 4 which is also formed as a cylinder casing.

10 Certain plastic substrates are decomposed at least superficially by titanium oxide layers. In the embodiment according to figure 5 therefore a protective layer 7 is deposited directly on the substrate 1, where this protective layer 7 has a thickness also lying in the nanometre range. This protective layer 7 is also applied on both sides consists of at least one metal oxide, preferably of the group ZnO, MgO, ZrO₂, In₂O₃, Sb₂O₃, Al₂O₃ and/or SiO₂, or a polar adhesion layer, for example, a polar plasma layer which also ensures good adhesion to the substrate 1,

Table 1

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The coating techniques and process parameters are adapted to the requirements for the product to be produced or substrate to be coated. Table 1 shows the production of selected functional titanium oxide layers and their protective and/or function effect. In the base layer 3 a relatively high content of hydrogen was analysed with ERDA (Elastic Recoil Detection Analysis), which is bonded in the layer in the form of hydroxide ions and depends on the process parameters and substrate temperature.

Composite materials which are coated on both sides were each given the same coating. During the coating process the substrate temperature is < 200°C. The anti-fogging effect is observed at a surface tension of > 50 mN/m and a correspondingly smooth surface. The surface tension also depends on the process parameters in production of the layer.

The thermal capacity of the fabric-like substrate which is coated increases virtually linear with the increasing layer thickness. According to the coated surface, the effect is visibly greater for PET film than for PET fabric. The thicker fabric mixture comprising 36% polyester and 64% viscose C shows a far less pronounced effect than the fine PET fabric. It is clear from this data that the layer thickness must be adapted to the substrate concerned (material, texture, thickness) in order to achieve the desired effect.

The average flame propagation speed for general textiles should be less than 90 mm/s, for textile curtains less than 60 mm/s. Even with a 12 nm thin ceramic coating, for a fine PET fabric the flame propagation speed lies far below the limit value of 60 mm/s and at a layer thickness of 180 nm achieves a value of 31 mm/s. For the viscose/polyester mixture there is a significant reduction from 142 to 115 mm/s with a 95 nm thick TiO_x/TiO₂ layer.

Table 1: Examples of processes for production of selected ceramic metal oxide layers with multiple protective or function effects

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Comments		Material layer	Base layer 3	Top layer 4	Material layer	Base laver 3		Intermediate layer 5	Top layer 4	Base layer 3	Ton lover A	lop layer 4	Material laver		Base layer 3	Intermediate layer 5	l op layer 4	Material layer	Base laver 3		Top layer 4	
Trans-	[%]				1 1 1 1 1 1				25				19	<u> </u>				1 1 1 1 1 1 1				
BIF d					! !								93	3				57		-		
Flame	spēed [mm/s]				55								744	115		-						
a [6	_				9.0	5							3.7	•				<u>; · </u>	╢			
[vol%]					2	2				-			8 5	? . <u>-</u>	_		 .	!	┨_			
Spec. thermal capacity	ŊĠ/ŢŌ-				1.6	F							2.4	4.				3.3				
Stoichio- metry	structure a		TiO _{1.9}	TiO ₂	1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1	TiO _{1.7} (OH) _{0.4}		TiO _{0.9}	TiO ₂	TiO _{21.9}	Ć	2	1 1 1 1 1 1 1 1 1		MgO	TiO _{1.0}	<u>5</u>	! ! ! ! ! ! !	TiO ₁₉		TiO ₂	
Thickness			12/180		2 × 12	360	·	75	20	70	25	3	2 × 95 2 × 95	2 x 95	80	15	,	2×100 1×100	40		350	
P ₍₀₂₎ /	[%]		15	06		10		7.5	25	7.5	02	2		•	0	00	>		8		09	
Process pressure	[hpar]		10	200	1	7		12	70	15	23	3	1		15	2,8	2	t 1 1 1 1 1	1000		2000	
Power			2000	800		800		1000	1000	009	1000	2			1500	700	3	. ————————————————————————————————————	1900		2700	
Process			• reactive DC sputtering	plasma post-oxidization	85 µm thick substrate	reactive DC sputtering	of TI(s) + O ₂	• reactive DC sputtering	or 11(s) + U ₂ • reactive DC sputtering of TI(s) + O ₂	reactive RF sputtering	of TI(s) + O ₂	of TI(s) + O ₂	85 µm thick substrate	64%/36% mixture	RF sputtering of MgO	RF sputtering of TiO(s)	• KF sputtering of TIO ₂ (s)	85 μm thick substrate	Plasma MO-CVD with	Ti(O-CH(CH ₃) ₂) ₄	Plasma MO-CVD with Ti(O-CH(CH ₃)) and	Fe(C _c H ₂) ₂) ₂
Layer		Substrate	۵		PET fabric		8			,	ပ		PET fabric PET film	Viscose/ PET		۵		PET fabric PET film		ш		

Legend

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- a. The stoichiometry of the layers and the layer surface was determined with RBS (Rutherford Backscattering Spectroscopy), ERDA (Elastic Recoil Detection Analysis) and XPS (X-ray Photoelectron Spectroscopy). The crystal structure of the layers was analysed qualitatively with TEM (Transmission Electron Spectroscopy) and XRD (X-ray). In the mixtures of amorphous and various crystalline phases (anatase, rutile and suboxides TiO_x (0.5 \leq x < 2) the corresponding phases could be identified in each case.
- b. The LOI (Limiting Oxygen Index) ISO 4589-2/ASTM D2863-77 describes the increase in limiting oxygen content in a gas mixture in the vol.% for a flame to combust the coated material.
- c. Burning speed, which was performed according to test 4589-2/ASTM D2863-77 (left-hand column) and average flame propagation speed, which was performed according to burning test BS EN ISO 6941 (right-hand column).
 - d. The BIF (Barrier Improvement Factor) shows the factor by which the oxygen permeability (measured in [ccm/m²_d.bar]) according to ASTM D 3985-95 at 0% r.h. and 23°C) diminishes due to coating of 12 μ m thick PET film in comparison with the uncoated PET film (124 cm/m².d.bar).
 - e. A coated pre-radiated glass material is immersed in a 0.05 mmol watery methylene blue solution and irradiated with a UV lamp (2 mW/cm²). The transmission change in solution is measured after 96 hours in a spectrophotometer at a wavelength of 650 nm according to the Sinku-Riko PCC-1.

The electrical conductivity and electrical resistance of the intermediate layer 5 concerned is given in example 3. The electric resistance of a 100 nm thick TiO₂

layer is more than $2.10^5 \,\Omega$ cm.

Examples

Some examples are described below for the production of multifunctional titanium oxide layers. In each case the layer properties and the layer structure are adapted to the product requirements concerned.

Example 1: Reactive Magnetron Sputtering with Subsequent Post-10 Oxidization

The deposition of a titanium oxide layer 2 on any substrate 1 with a reactive sputtering process (DC = (pulsed) direct current; RF = radio frequency) of titanium with a mixture of process gases of argon and oxygen. Then by a change in plasma conditions (variant 1a) and/or with post-oxidization (variant 1b) of the composite material a TiO₂ top layer 4 containing anatase is formed.

*Coating process:

Target:

Titanium metal (99.98%)

20 Power:

1 - 7 W/cm² DC/RF

Process pressure:

10 μbar

Partial pressure $p(O_2)/p(tot)$:

10% DC/RF

Variant 1a: TiO₂ layer at the end of the process

In the last phase of the reactive sputtering process, the process pressure is increased to 20 μbar and in the case of the DC sputtering process the oxygen partial raised to 30%, in the case of RF sputtering process the oxygen partial pressure is raised to 60%. The increase in process pressure and oxygen partial pressure acts favourably on the layer properties of the top layer which are characterised by a lower density, a higher porosity and hence larger surface.

Variant 1b: Post-oxidization of TiO_x to TiO₂

In this case the titanium oxide layer is oxidized in an oxidizing atmosphere with a

PE-CVD at low pressure to atmospheric pressure. The penetration depth of the post-oxidization depends on the density of the TiO_x layer and the process conditions.

5 Power (pulsed/continuous):

50 - 3000 W radio frequency (MHz),

hyperfrequency (GHz) or low frequency (kHz)

Process pressure:

0.1 mbar - 1 bar

Partial pressure $p(O_2)/p(tot)$:

50 - 100%

10 Example 2: Adhesion-promoting pretreatment and plasma-activated MOCVD process

Plasma activation of a substrate take place (1) to increase the adhesion of the coating.

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Pretreatment:

Power (pulsed/continuous):

200 - 1500 W hyperfrequency (2.45 GHz)

Process pressure:

20 μbar - 1 bar

Partial pressure p(O₂/N₂O)/p(tot): 20 - 80%

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Base layer 4 comprising TiO_x or TiO_x mixed with SiO_x :

Then introduced into the reaction chamber is a titanium-containing monomer gas for example titanium tetrakis-isopropoxide (TTIP) (Ti(O-CH(CH₃)₂)₄) together with oxygen and one or more inert gases (Ar, He), and a TiO_x layer 4 deposited. In addition hexamethyldisiloxane (HMDSO) can be introduced into the plasma process so as to give a ratio of the two metal oxides in the base layer of 2:1.

Power (pulsed/continuous):

600 - 3500 W hyperfrequency (2.45 GHz)

Process pressure:

10 μbar - 0.1 bar

30 Process gases:

Ar/He as carrier gas through (Ti(O-CH(CH₃)₂)₄)

at 50°C, Ar/he and O₂.

Top layer 4 comprising TiO₂ or TiO₂ doped with Fe₂O₃:

Then on the base layer 3, after the titanium-containing process gases, as a carrier gas a small quantity of an iron-containing monomer gas is introduced into the reaction chamber (e.g. iron-acetylacetonate complex $Fe(C_5H_7O_2)_3$, with oxygen and one or more inert gases (Ar, He etc), to deposit an anatase-containing TiO_2 top layer 4 doped with 0.1 - 9 at% Fe_2O_3 . At the same time by varying the process parameters, the layer structure can be modified.

Using numerous energy-rich plasma-activated discharges from low frequency up to hyperfrequency range and combinations thereof it is possible to produce the composite materials described. Examples are (Remote) AP-PECVD (atmospheric pressure plasma-enhanced chemical vapour deposition), APNEP (atmospheric pressure non-equilibrium plasma), plasma jet, plasma broad beam burner, microwave discharge, pulsing surface discharge, DBD (dielectric barrier discharge), APGD (atmospheric pressure glow discharge).

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Example 3: Electrically conductive TiO_x intermediate layer 5

An electrically conductive intermediate layer 5 is produced which is more conductive than the base layer 3 and/or the additional base layer 7. The TiO_x layer $(0.7 \le x < 1.5)$ is deposited with any substrate 1 fitted with a base layer 3, in that in a reactive sputtering process less oxygen gas is supplied to the process than in the base layer 3 and the process pressure is adapted. It is also possible to deposit the TiO_x layer non-reactively in a sputtering process using a corresponding target (TiO_1 , Ti_2O_3 , Ti_3O_2 etc.).

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Reactive DC sputtering process to obtain a $TiO_{1.0}$ layer with an electrical resistance of 1.2.10⁻² Ω cm or 50 Ω cm:

Target:

Titanium metal (99.98%)

30 Power:

3 W/cm² DC

Process pressure:

20 μbar or 7 μbar

Partial pressure $p(O_2)/p(tot)$:

5% or 7.5%